

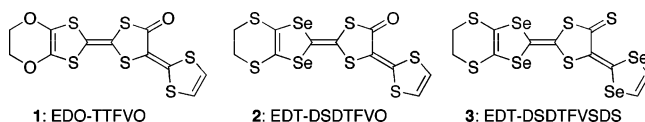
## A Metallic (EDT-DSDTFVSDS)<sub>2</sub>·FeBr<sub>4</sub> Salt: Antiferromagnetic Ordering of *d* Spins of FeBr<sub>4</sub><sup>-</sup> Ions and Anomalous Magnetoresistance Due to Preferential $\pi$ -*d* Interaction

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The crystals of cation radical salts of  $\pi$  donor molecules with magnetic transition metal counteranions usually form alternate stacking structures of the donor and counteranion layers, which are responsible for electrical conductivities and magnetisms of the crystals, respectively.<sup>1,2</sup> Current interest is directed toward the coexistence of metallic conductivity and ferromagnetism by virtue of a significant  $\pi$ -*d* or  $\pi$ -*f* interaction between the metal-conducting  $\pi$  electrons on the donor layers and the localized *d* or *f* spins on the counteranion layers. Such conducting  $\pi$  electrons are spin-polarized by the magnetic layers, and can be utilized as a key component for developing a new type of molecular electronics, called "spin electronics or spintronics". In the past decade, considerable efforts have been devoted to the development of a ferromagnetic molecular metal with the  $\pi$ -*d* interaction, but it has not been obtained yet. Although metallic conductivities have been realized in  $\kappa$ -(BETS)<sub>2</sub>·FeX<sub>4</sub> [BETS = bis(ethylenedithio)tetraselenafulvalene, X = Cl, Br],<sup>3</sup> (EDO-TTFVO),<sup>1</sup> (1)<sub>2</sub>·FeCl<sub>4</sub>,<sup>4</sup> and (EDT-DSDTFVO),<sup>2</sup> (2)<sub>2</sub>·FeX<sub>4</sub> (X = Cl, Br),<sup>5</sup> the Fe(III) *d* spins of these FeX<sub>4</sub><sup>-</sup> salts were antiferromagnetically ordered except for 2<sub>2</sub>·FeX<sub>4</sub>. The main origin of the antiferromagnetic ordering in these salts is due to a *d*-*d* interaction between the FeX<sub>4</sub><sup>-</sup> ions in the anion layer in preference to a  $\pi$ -*d* interaction between the donor molecules and FeX<sub>4</sub><sup>-</sup> ions. Because such a *d*-*d* interaction is usually antiferromagnetic, the antiferromagnetic ordering cannot be avoided as far as it is dominant compared with the  $\pi$ -*d* interaction. In contrast, there is a possibility that the  $\pi$ -*d* interaction brings about the ferromagnetic ordering, which will be discussed here. For the first time, we could find out a preferential  $\pi$ -*d* interaction in the 2:1 salt of a newly prepared donor molecule, EDT-DSDTFVSDS (**3**) with a magnetic FeBr<sub>4</sub><sup>-</sup> ion, 3<sub>2</sub>·FeBr<sub>4</sub>, which is essentially metallic down to 4.2 K despite a small upturn in resistivity below ca. 30 K and shows an antiferromagnetic ordering of the *d*-spins at *T*<sub>N</sub> = 3.3 K.



According to a method similar to that used to synthesize 2,<sup>5</sup> **3** was obtained from 4,5-ethylenedithio-4',5'-bis(cyanoethylthio)-

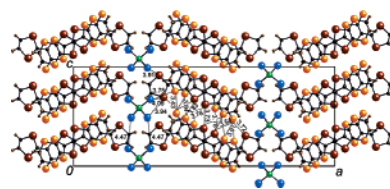


Figure 1. Crystal structure of 3<sub>2</sub>·FeBr<sub>4</sub> projected down to the *ac*-plane.

diselenadithiafulvalene and 2-methylseleno-1,3-diselenolium tetrafluoroborate.<sup>6</sup> MO calculations suggest that the HOMO of **3** has large atomic coefficients on the Se atoms of the terminal 1,3-diselenole ring. The increased coefficients on the Se atoms can play an important role to produce the preferential  $\pi$ -*d* interaction through the Se...halogen contacts together with the bulkiness of the 1,3-diselenole ring which might weaken the *d*-*d* interaction.

A solution of **3** and 10 equiv of *n*-Bu<sub>4</sub>N<sup>+</sup>MBr<sub>4</sub><sup>-</sup> (M = Fe, Ga) in chlorobenzene/ethanol (9:1, *v/v*) was electrochemically oxidized with a constant current of 0.1  $\mu$ A at 45 °C. Black platelike crystals were grown on the anode after several weeks. The crystal structures of 3<sub>2</sub>·FeBr<sub>4</sub> and 3<sub>2</sub>·GaBr<sub>4</sub> were successfully solved and isostructural to each other.<sup>7</sup> Each crystal involved one crystallographically independent donor molecule with an almost planar molecular structure. As seen from the projection down to the *ac*-plane (Figure 1), the donor molecules form uniformed stacking columns with an equal interplanar distance of 3.75 Å and several S...S and S...Se contacts along the *b*-axis, making each terminal 1,3-diselenole ring projected into the outside. Furthermore, these stacking columns are aligned along the *c*-axis, and the donor layer has a  $\beta$ -like packing motif. Several close S...Se contacts are also observed between neighboring columns, suggesting a strong two-dimensional intercolumnar interaction. On the other hand, the FeBr<sub>4</sub><sup>-</sup> ions intervene between the donor layers and are aligned in a rectangular form in the *bc*-plane, but neighboring FeBr<sub>4</sub><sup>-</sup> ions are largely separated from each other by the projected 1,3-diselenole rings. Accordingly, the shortest Br...Br distances between neighboring FeBr<sub>4</sub><sup>-</sup> ions is 4.47 Å, which is very much longer than van der Waals distance (3.90 Å). In contrast, the FeBr<sub>4</sub><sup>-</sup> ion has very close Br...S (3.75 Å) and Br...Se (3.91, 3.94, 4.06, and 4.13 Å) contacts with neighboring donor molecules. These structural features suggest a stronger  $\pi$ -*d* interaction between the FeBr<sub>4</sub><sup>-</sup> ions and the donor molecules than the *d*-*d* interaction.

The band calculation of 3<sub>2</sub>·FeBr<sub>4</sub> was performed by a tight-binding method based on the extended Hückel approximation. The overlap integral along the stacking direction (*b* = -39.71  $\times 10^{-3}$ ) is four to six times larger than those along the side-by-side direction (*p* =

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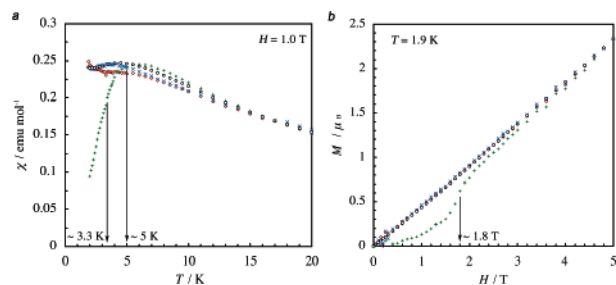
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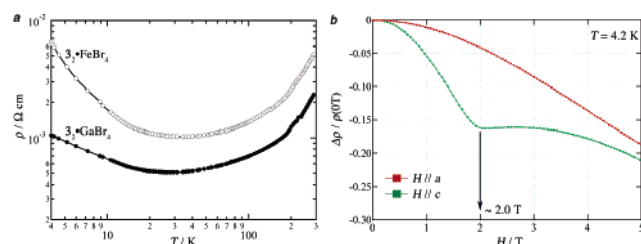
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**Figure 2.** (a)  $\chi$  vs  $T$  ( $H = 1.0$  T,  $T < 20$  K) and (b)  $M$  vs  $H$  ( $T = 1.9$  K,  $H < 5$  T) for  $H//a$  (red diamond),  $H//b$  (blue cross) and  $H//c$  (green plus) directions of the single crystal and for the polycrystalline sample (open circles) of  $3_2\text{-FeBr}_4$ .



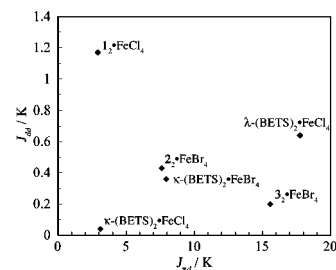
**Figure 3.** (a) Temperature dependences of  $\rho$  values in the temperature range of 4.2 to 290 K for  $3_2\text{-FeBr}_4$  (open circles) and  $3_2\text{-GaBr}_4$  (closed circles); (b) MR effect [ $\Delta\rho/\rho(0$  T)] for  $3_2\text{-FeBr}_4$  in the  $H//a$  (red) and  $H//c$  (green) directions in the  $H$  range of  $<5$  T at 4.2 K.

$9.44 \times 10^{-3}$  and  $q = 7.11 \times 10^{-3}$ ), suggesting a quasi-one-dimensional electronic structure. The calculated Fermi surface has a two-dimensionality, but opens along the  $k_c$ -direction similar to that of  $2_2\text{-FeBr}_4$ .<sup>5</sup>

Magnetizations ( $M$ ) of  $3_2\text{-FeBr}_4$  were measured on a single crystal using a SQUID magnetometer under  $H = 1.0$  T applied along the three crystallographic axes ( $a$ -,  $b$ -, and  $c$ -axes). The temperature dependence of magnetic susceptibilities ( $\chi$ ) measured on a polycrystalline sample can be fitted to the Curie–Weiss law with a Curie constant of  $4.71$  emu K mol<sup>-1</sup> and a Weiss temperature of  $-10.5$  K, suggesting a relatively strong antiferromagnetic interaction between the  $d$  spins of the  $\text{FeBr}_4^-$  ions. As shown in Figure 2a,  $\chi$  sharply decreased below ca. 5 K only for  $H//c$ , suggesting an antiferromagnetic ordering. The accurate Néel temperature ( $T_N$ ) was determined to be 3.3 K by a heat capacity measurement. The  $M$ – $H$  curves measured at 1.9 K showed a spin-flop around 1.8 T for  $H//c$  (see Figure 2b). These results indicate an antiferromagnetic ordering at 3.3 K with an easy axis parallel to the  $c$ -axis.

Electrical resistivities ( $\rho$ ) of the single crystals of  $3_2\text{-FeBr}_4$  and  $3_2\text{-GaBr}_4$  were measured down to 4.2 K. The electrical conductivities of  $3_2\text{-FeBr}_4$  and  $3_2\text{-GaBr}_4$  measured along the  $b$ -axis are 200–300 S cm<sup>-1</sup> at 290 K. As shown in Figure 3a, essentially metallic behaviors are observed despite small upturns in  $\rho$  below ca. 30 K, and the  $\sigma$  values at 4.2 K are still very high (ca. 170 and 1000 S cm<sup>-1</sup>, respectively). Below ca. 30 K, the increasing degree in  $\rho$  is larger for the  $\text{FeBr}_4^-$  salt than for the  $\text{GaBr}_4^-$  salt. The magnetoresistance (MR) effect [ $\Delta\rho/\rho(0$  T)] was investigated under  $H \perp b$  ( $H//ac$ -plane). For the  $\text{GaBr}_4^-$  salt, a very small positive MR effect (ca. 3% at 5 T) was observed at 4.2 K. In contrast, the  $\text{FeBr}_4^-$  salt exhibited a large and negative MR effect (ca. 20% at 5 T) as shown in Figure 3b. Very interestingly, an anomalous MR with a broad dip appeared at ca. 2.0 T for  $H//c$  without hysteresis, but was not observed for  $H//a$ .<sup>8</sup> This dip is clearly responsible for the spin-flop transition of the  $d$  spins of the  $\text{FeBr}_4^-$  ion which occurs at almost the same  $H$  in the  $c$ -direction (easy axis) and gives a definite evidence for the strong  $\pi$ – $d$  interaction.

The magnitudes of the  $d$ – $d$  ( $J_{dd}$ ) and  $\pi$ – $d$  ( $J_{\pi d}$ ) interactions in  $3_2\text{-FeBr}_4$  were estimated to be 0.20 and 15.6 K, respectively, by



**Figure 4.** Calculated  $J_{dd}$  and  $J_{\pi d}$  values for  $1_2\text{-FeCl}_4$ ,  $2_2\text{-FeBr}_4$ ,  $3_2\text{-FeBr}_4$ ,  $\kappa\text{-(BETS)}_2\text{-FeCl}_4$ ,  $\kappa\text{-(BETS)}_2\text{-FeBr}_4$ , and  $\lambda\text{-(BETS)}_2\text{-FeCl}_4$ .

the MO calculations.<sup>9</sup> Figure 4 shows the comparison of these  $J_{dd}$  and  $J_{\pi d}$  values with those in molecular metallic conductors with  $\text{FeX}_4^-$  ( $X = \text{Cl}, \text{Br}$ ) ions so far known,  $1_2\text{-FeCl}_4$ ,<sup>4</sup>  $2_2\text{-FeBr}_4$ ,<sup>5</sup>  $\kappa\text{-(BETS)}_2\text{-FeCl}_4$ ,  $\kappa\text{-(BETS)}_2\text{-FeBr}_4$ , and  $\lambda\text{-(BETS)}_2\text{-FeCl}_4$ .<sup>9</sup> The  $J_{\pi d}$  in  $3_2\text{-FeBr}_4$  is very large and comparable to that in  $\lambda\text{-(BETS)}_2\text{-FeCl}_4$ , which exhibited novel conducting and magnetic behaviors such as a field-induced superconductivity as a result of a strong  $\pi$ – $d$  interaction,<sup>10</sup> while the  $J_{dd}$  is very small in comparison to that in  $\lambda\text{-(BETS)}_2\text{-FeCl}_4$ , suggesting that the  $\pi$ – $d$  interaction is dominant in  $3_2\text{-FeBr}_4$ . Accordingly,  $3_2\text{-FeBr}_4$  is expected to be the most reliable  $\pi$ – $d$  system to produce a ferromagnetic ordering of the  $d$  spins of the  $\text{FeBr}_4^-$  ions through the strong  $\pi$ – $d$  interaction with metal-conducting  $\pi$  electrons. However, the antiferromagnetic ordering was achieved at  $T_N = 3.3$  K. Nevertheless, the present result encourages us to develop a novel ferromagnetic molecular metal with a significant  $\pi$ – $d$  interaction by use of new derivatives of **3**.

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**Supporting Information Available:** An X-ray crystallographic file for  $3_2\text{-FeBr}_4$  and  $3_2\text{-GaBr}_4$  in the CIF format, as well as the results of HOMO of **2** and **3**, crystal structure, band structure, and estimation of  $J$  values of  $3_2\text{-FeBr}_4$  in the PDF format. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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- Compound **3**: a purplish black microcrystals; mp 199–200 °C (dec). <sup>1</sup>H NMR (400 MHz, *d*<sub>6</sub>-acetone):  $\delta$  8.07 (*d*,  $J = 6.8$  Hz, 1H), 7.98 (*d*,  $J = 6.8$  Hz, 1H), 3.32 (s, 4H). Anal. Calcd for  $\text{C}_{11}\text{H}_6\text{Br}_2\text{Fe}_0.5\text{S}_2\text{Se}_4$ : C, 21.51; H, 0.98. Found: C, 21.33; H, 1.05%. MS (EI) *m/z*: 614 ( $M^+$ ).
- Crystal data for  $3_2\text{-FeBr}_4$ :  $\text{C}_{11}\text{H}_6\text{Br}_2\text{Fe}_0.5\text{S}_2\text{Se}_4$ ,  $M_r = 802.04$ , orthorhombic,  $a = 37.64(4)$ ,  $b = 7.498(6)$ , and  $c = 14.03(1)$  Å,  $V = 3960(6)$  Å<sup>3</sup>,  $T = 296$  K, space group  $Pccn$ ,  $Z = 8$ ,  $\mu(\text{Mo K}\alpha) = 12.333$  cm<sup>-1</sup>, 48066 reflections measured ( $R_{\text{int}} = 0.100$ ), 4305 unique, of which 1968 were used in all calculations [ $F^2 > 1.00\sigma(F^2)$ ]. The final  $R$  and  $R_w$  were 0.057 and 0.095, respectively. Crystal data for  $3_2\text{-GaBr}_4$ :  $\text{C}_{11}\text{H}_6\text{Br}_2\text{Ga}_0.5\text{S}_2\text{Se}_4$ ,  $M_r = 808.98$ , orthorhombic,  $a = 37.766(6)$ ,  $b = 7.5021(11)$ ,  $c = 14.043(2)$  Å,  $V = 3978.7(10)$  Å<sup>3</sup>,  $T = 293$  K, space group  $Pccn$ ,  $Z = 8$ ,  $\mu(\text{Mo K}\alpha) = 12.586$  cm<sup>-1</sup>, 53826 reflections measured ( $R_{\text{int}} = 0.048$ ), 5124 unique, of which 2677 were used in all calculations [ $F^2 > 3.00\sigma(F^2)$ ]. The final  $R$  and  $R_w$  were 0.054 and 0.066, respectively.
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